In situ ATR-FTIR and SFM Studies on the Influence of Adsorption time on Deposition and Nanostructure of Poly(ethyleneimine)/Poly(acrylic acid) Multilayers

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Summary: The deposition and the nanostructure of polyelectrolyte multilayers (PEM) of branched poly(ethyleneimine)/poly(acrylic acid) (PEI/PAC) was studied in dependence of the adsorption time (t_{ADS}) of the individual steps. PEM were reproducibly deposited applying up to z=20 adsorption steps at the fixed pH combination of 10/4 and polyelectrolyte concentration $c_{PEL}=0.005$ M in a flow cell using an automated valve system. in situ ATR-FTIR spectroscopy and SFM were used for quantitative determination of deposited amount and thickness, respectively. A linear relation between PEL band integrals and thickness of thin PEM films was found. Varying t_{ADS} from 0.5 to 5 min in each of the adsorption steps resulted in a steep rise of the deposited PEM amount. For $t_{ADS} > 5$ min the deposition did only marginally increase. Evidence for the release of outermost located PEI upon PAC immersion (even step) and of outermost PAC upon PEI immersion (odd step) was obtained. SFM images on consecutively deposited PEM-6 showed a slight increase in structure size and roughness for increasing t_{ADS} . These studies help to prepare polyelectrolyte based films with controlled thickness for the interaction with biofluids in the biomedical and food field.

Keywords: ATR-FTIR; deposition; poly(acrylic acid); poly(ethyleneimine); polyelectrolyte multilayers; SFM

Introduction

Polyelectrolyte (PEL) multilayers (PEM) can be used as a simple surface modification technique for curved and planar substrates based on aqueous systems. PEM were introduced by Decher^[1] in the early nineties and have become an interesting topic of colloid and surface science, which is therein.^[2–4] reviewed comprehensively PEM have wide application fields in biomedicine,[5-8] diagnostics and sensorics, [9-11] and separation technology. [12-14] In principle, the PEM deposition process is based on the consecutive adsorption of polycations (PC) and polyanions (PA), typically on charged substrates beginning

with the oppositely charged PEL. Although the preparation of PEM can be easily performed, fundamental issues like overcompensation and growth mechanism, location of the counterions, internal PEL order and composition, surface morphology and longterm stability of PEM are still not completely resolved. Concerning PEM growth mechanism the traditional picture is based on the linear relation between the UV absorbance of one PEM component like poly(styrene sulfonate) and the bilayer number, from which a well defined regular PEL uptake of a constant thickness increment was derived.[1] From the beginning of 2000 this picture was partly reviewed by the observation of exponential growth in certain systems, especially when charged polypeptides like poly(L-lysine) or poly (L-glutamic acid) were involved, [15,16] which was explained by diffusion of PEL "in" and "out" of the PEM. A three zone model of

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PEM^[17] was propagated claiming a non-stoichometric zones for the surface (I) and the outermost region (III) and a 1:1 stoichometry in the core zone II between I and III. Recently, the PEM deposition process was found to consist of both uptake of the incoming immersed PEL and rupture of the previously adsorbed PEL by the respective oppositely charged PEL, as it was convincingly shown by Cohen-Stuart.^[18] While they used reflectometry to show that upon immersing the new PEL solution an initial overall thickness decrease occurs, systematic studies are lacking on molecular composition changes.

In this paper we focus on the deposition and molecular composition of a PEM system consisting of commercially available branched poly(ethyleneimine) (PEI) and poly(acrylic acid) (PAC) using quantitative in situ attenuated total reflexion Fourier transform infrared (ATR-FTIR) spectroscopy and scanning force microscopy (SFM). While accumulated knowledge from the literature is available on the influence of parameters like chain charge, [19] ionic strength^[20-22] and pH,^[23,24] on PEM deposition the influence of adsorption time has not been systematically studied yet. Herein the influence of the adsorption time in every individual adsorption step is focussed at. Therefore, as main parameter the adsorption time of each of the respective deposition steps was varied, while pH and PEL concentration were kept constant.

Experimental Part

Materials

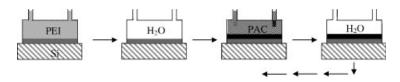
Commercial branched poly(ethyleneimine) (PEI, 750.000 g/mol, branching degree:

65%^[25]) from Sigma-Aldrich (Steinheim, Germany) and linear poly(acrylic acid) (PAC, 50.000 g/mol) from Polysciences Inc. (Warrington, U.S.A.) were used. Polyelectrolyte (PEL) solutions were prepared by dissolving in Millipore water at concentrations of $c_{PEL} = 0.005 \text{ M}$. The pH values of the unbuffered polyelectrolyte solutions were 10.0 ± 0.1 for PEI and 4.0 ± 0.2 for PAC solution, respectively. Trapezoidal silicon internal reflection elements (IRE, $50 \times 20 \times 2 \text{ mm}^3$) were purchased from Komlas GmbH (Berlin, Germany), which were at first placed in a mixture of H_2SO_4/H_2O_2 (50/50 v/v%), then in water, alcohol and chloroform and finally were cleaned by UV plasma under reduced pressure (plasma cleaner/sterilizer PDC-32 G, Harrick, Ossining, NY, U.S.A.) to remove organic impurities and create reproducible surface properties.

PEM Depositon

PEM were prepared by injecting consecutively in that sequence PEI solution at pH=10, pure water, PAC solution at pH=4 at a concentration of $c_{PEL}=0.005$ M into the S compartment (see below) of the *in-situ-*ATR cell as it is shown in the Figure 1.

An automated valve system^[26] (M.M., W. Jenschke, IPF Dresden e.V., Wünschmann GmbH, Dresden) triggering the IR spectroscopic measurements was used, by which flow (mL/min) and adsorption time (min) could be varied according to the following protocol: (i) Injection of PEI solution at a flow of 40 mL/min for 10 sec, then further adsorption at a flow of 1 mL/min for 5 min, (ii) injection of Millipore water (flow: 40 mL/min) for 10 sec, then further rinsing (flow: 1 mL/min) for 150 sec



Polyelectrolyte multilayer (PEM) deposition in the *in-situ* ATR cell (grey: polycation solution (PC), black: polyanion solution (PA), white: Millipore water (H₂O).

and IR measurement, (iii) injection of PAC solution (flow: 40 mL/min) for 10 sec, then further adsorption (flow: 1 mL/min) for 5 min, (iv) injection of Millipore water (flow: 40 mL/min) for 10 sec, then further rinsing (flow: 1 mL/min) for 150 sec and IR measurement, (v) = (i) (iterations).

Scanning Force Microscopy

SFM measurements were performed on the Ultramiscroscope consisting of an optical microscope and SFM attachment (Nanostation II, SIS GmbH, Herzogenrath, Germany). Silicon probe tips from Nanosensors (Darmstadt, Germany) were used having radii of around 10 nm. The measurements were performed in the "noncontact mode" directly on the dry PEM samples on Si-IRE (internal reflection elements) used in the ATR-FTIR measurements under room conditions. The SFM images were recorded in topography, error and phase mode. As soon as artefacts (e.g. triangles as convolution of the tip with the object) appeared, the tip was immediately replaced. The scanning parameters were optimized by minimizing the signal in the error mode. Surface profiles were generated from SFM raw data by the SISCANPro software. The RMS roughness values were determined from the respective SFM images in the topography mode. Film thickness was measured based on topography images $(32 \times 32 \mu m)$ of zones treated by careful scalpel cuts considering the height difference between undamaged film and bare silicon.

in situ ATR-FTIR Spectroscopy

The principle of in situ attenuated total Fourier transform infrared (ATR-FTIR) spectroscopy is given in the Figure 2. ATR-FTIR measurements were performed on the in situ ATR-FTIR apparatus (Optispec, Zürich, Switzerland), which was installed on a FTIR spectrometer (IFS 55, Bruker Optik GmbH, Leipzig, Germany). This attachment consists of a special mirror setup and a transparent in situ sorption cell (M.M., IPF Dresden e.V.) using a plasma cleaned silicon internal reflection element (Si-IRE). ATR-FTIR absorbance spectra were recorded by the single-beam-samplereference (SBSR) technique^[27] based on probing separately the upper sample (S) and lower reference (R) half of a trapezoidal Si-IRE $(50 \times 20 \times 2 \text{ mm}^3, \text{ N} = 11 \text{ active})$ reflections on the shorter side), which was sealed by O-rings and front and back side of the in-situ cell, by a single IR beam. Wavenumber dependent intensities recorded for $S I_S(v)$ were devided by those for R $I_R(\nu)$ and the absorbance was computed by $A_{SBSR} = -\log (I_S(v)/I_R(v))$. Typically, the in situ ATR-FTIR measurements on the consecutive deposition of PEI/PAC were triggered on-line by the automated

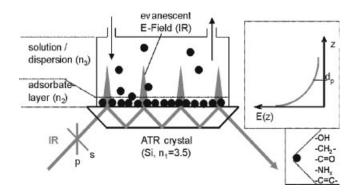


Figure 2.Principle of in-situ ATR-FTIR spectroscopy featuring exponentially decaying evanescent waves at the optically dense/rare medium interface to probe surface attached organic material. (From [28] with kind permission).

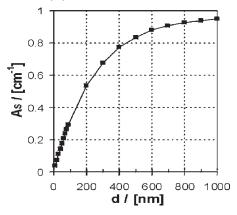


Figure 3. Course of the ATR measured absorbance A_S in dependence of the variable layer thickness d according to Equation (1) and (2) using arbitrary constant values for N, ε and c. (From^[28] with kind permission).

valve system. ATR-FTIR spectra (50 scans were coadded) were always recorded after rinsing the respective PEI or PAC solution out of the sample compartment by Millipore water (see protocol above). The spectral resolution was 2 cm⁻¹. No significant spectroscopic changes were observed comparing ATR-FTIR spectra recorded in the presence of PEI or PAC solution with those recorded after rinsing and in the presence of Millipore water.

Processing of the ATR-FTIR Data

In the ATR-FTIR spectra on (especially thick) PEM films in liquid water at the silicon IRE surface a negative $\nu(OH)$ stretching band at around 3380 cm⁻¹ as well as a negative $\delta(OH)$ band at 1643 cm⁻¹ appeared shown in the Figure 4a, which is due to the lower concentration or removal of liquid water at the modified IRE surface, as it will be pointed out later. (Other spectra in Figure 4a are explained below).

Since especially the negative $\delta(OH)$ band overlaps the diagnostic $\nu(C=O)$ and $\nu(COO^-)$ bands of poly(acrylic acid) (see below), this band has to be taken into account for an exact data analysis. This problem was addressed by lineshape analysis (LSA) in the range from 2000–1250 cm⁻¹,

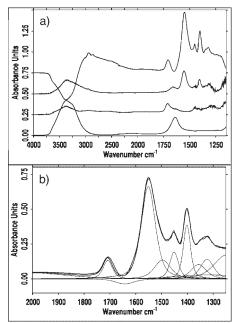


Figure 4. a) ATR-FTIR spectra of liquid water (scaled by 0.5), 1 M PAC at pH = 2, 1 M PAC at pH = 10 and PEM-10 in contact to Si-IRE (bottom to top). b) Lineshape analysis results on a typical PEM spectrum (PEM-10) using nine positive and one negative Lorentzian/ Gaussian (50/50%) component at 1643 cm $^{-1}$ (δ (OH)) (constraint fit).

representing the lineshape by the superposition of ten Lorentzian/Gaussian (50%/ 50%) bands. At first LSA was performed under free relaxation of the parameters (wavenumber) position, halfwidth and intensity for nine components and all PEM-z spectra (z=2 to 20). As tenth component a negative $\delta(OH)$ band at 1643 cm⁻¹ with a halfwidth of 85 cm⁻¹, whose parameters were found by fitting a bulk water spectrum (Figure 4a, bottom spectrum), was kept fixed and only the (negative) intensity was relaxed. This procedure evidenced nine further positive components at 1250, 1323, 1354, 1402, 1451, 1490, 1552, 1707 and 2000 cm⁻¹, which could be also identified by visual inspection. Note that the 1250 and 2000 cm⁻¹ components were used as "dummy" components to avoid boundary effects on the intensities

of the critical 1707, 1643 and 1552 cm⁻¹ components. Then the respective positions and halfwidths of all peaks were averaged and in a second run all these spectra were fitted fixing the averaged values of position and halfwidth and relaxing solely the intensities (constraint fit). A representative constraint fit result is shown for PEM-10 in the Figure 4b, featuring the original and simulated lineshape and the ten overlapping components.

Quantitative Analysis of the ATR-FTIR Data

Exact quantitative ATR-FTIR analysis is based on a modified Lambert-Beer law given in Equation (1):

$$A_{S} = N \varepsilon c d_{E,S}$$
 (1)

including the integrated absorbance of a given IR band measured in s-polarization A_S [cm $^{-1}$], number of active reflections N, absorption coefficient ϵ [cm/Mol], concentration c [Mol/cm 3], and the effective thickness $d_{E,~S}$ [cm $^{-1}$] due to Harrick, [29] which is given by:

$$\begin{split} d_{E,S} &= n_2 \; d_p \; E_Y^2/(2\cos\theta \; n_1) \\ &* [1 - \exp(-2d/d_p)] \end{split} \tag{2}$$

d_{E, S} is a function of the refractive indices of the involved media n₁ (Si, 3.5), n₂ (PEM, 1.5) and n₃ (water, 1.33), incident angle of IR beam θ (45°), thickness d and the >penetration depth d_P (\approx 260 nm at 3000 cm⁻¹, 460 nm at 1700 cm⁻¹ and 510 nm at 1550 cm⁻¹) and the relative electrical field component in y-direction E_Y = $2\cos\theta/(1-(n_3/n_1)^2)^{1/2}$. The surface concentration Γ [Mol/cm²] can be obtained from by inserting c from (1) into following Equation (3):

$$\Gamma = c d \tag{3}$$

A typical course of the measured absorbance A_S of an organic layer system in dependence of its thickness d is given in the Figure 3. Two points are characteristic: At first for thin films between d=0 up to 300 nm a quasi linear dependence of A_S on d can be approximated. Secondly, for thick

films A_S runs into a saturation, meaning that the ATR-FTIR method gets increasingly insensitive to additional layer deposition or the nature of outer film regions, when d exceeds penetration depth dp ≈ 500 nm. For the semiquantitative determination of Γ the integrated areas of typical polyelectrolyte (PEI, PAC) IR bands A, which were measured by unpolarized IR beam, were considered. For the determination of PEI surface concentration the integral of the $\nu(CH)$ band at 2855 cm⁻¹ (limits: 2890–2820 cm⁻¹) was used denoted as A_{CH} (= A_{PEI}). For the PAC surface concentration the integral of the $\nu(C=O)$ component $(A_{C=O})$ at 1707 cm⁻¹ and of the $\nu(COO^-)$ component (A_{COO^-}) at 1552 cm⁻¹ were used according to the relation $A_{PAC} = F \cdot A_{C=O} + A_{COO}$. In that work F = 2.24 was used, which is due to the ratio of the absorption coefficients of $\nu(COO^-)$ and $\nu(C=O)$ band. This ratio can be obtained from the spectra of 1 M PAC at pH=2 (only ν (C=O)) and at pH = 10 (only $\nu(COO^{-})$), respectively, which are given in the Figure 4a (above). These integrals A_{PEI} and A_{PAC} are approximately proportional to the surface concentration (denoted further as deposited amount) of PEI and PAC, respectively, if the overall thickness of the PEM film d < 300 nm (see Figure 3). Otherwise for d > 300 nm the PEL deposited amount is increasingly underestimated for increasing PEM thicknesses. For relative comparisons normalized values of A_{PEI} and A_{PAC} were used, whereby the actual APEI or APAC value was devided by the corresponding value obtained for an adsorption step $z \gg 10$, where the respective band integral was in saturation (no further signal increase in dependence of z).

Results and Discussion

In the following results on the characterization of consecutively adsorbed PEM consisting of branched PEI and PAC at silicon IRE at a fixed pH setting and constant c_{PEL}, which were obtained by

ATR-FTIR spectroscopy and SFM, will be presented. The main parameter studied was the adsorption time $t_{\rm ADS}$. Branched PEI was chosen, since it was expected to result in higher deposited amounts compared to linear PEI. ATR-FTIR spectroscopy was used to determine deposited amount and chemical composition in dependence of $t_{\rm ADS}$, while SFM performed on the samples previously measured by ATR-FTIR gave informations on the formed nanostructures in dependence of $t_{\rm ADS}$.

The paper is structured as follows: At first typical *in-situ*-ATR-FTIR spectra on consecutive PEI/PAC deposition in dependence of the adsorption step z are presented and assignments of typical IR bands related selectively to PEI and PAC, respectively, are given. Secondly, deposition profiles, i.e. courses of typical band integrals in dependence of t_{ADS} and interpretations of their non trivial courses are given. Thirdly, SFM images on equivalent samples previously characterized by ATR-FTIR are presented illustrating the influence of t_{ADS} on PEM nanostructure. Finally, the results are discussed and conclusively summarized.

ATR-FTIR Spectra

Typical ATR-FTIR spectra on the consecutive deposition of PEI and PAC on Si-IRE are given in the upper part of

Figure 5 for a single PEI layer (z=1, PEM-1, bottom) up to twelve consecutively adsorbed PEI and PAC layers (z=12, PEM-12).

As fixed parameters settings $c_{PEL} =$ 0.005 M and $pH \approx 10 \text{ (PEI)}$ and $\approx 4 \text{ (PAC)}$ were chosen, since under these conditions the deposited amount was found to be highest [30]. In these original uncorrected PEM spectra the increasing overall intensity and the changes of both the $\nu(C=O)$ band (carbonyl stretch) at around 1707 cm⁻¹ and the $\nu(COO^-)$ band (asymmetric carboxylate stretch) at 1552 cm⁻¹, respectively, in dependence of z are most significant, which both can be exclusively attributed to the PAC component. Moreover, a weak $\nu(CH)$ band between 2890–2820 cm⁻¹ is important, which can be exclusively assigned to the CH and CH₂ moieties of PEI (Figure 5, right part). This was recently confirmed by ATR-FTIR spectra on bulk 1 M PEI compared to 1 M PAC solution therein.^[30] Furthermore, an increasing negative v(OH) band at around 3380 cm⁻¹ shows up, which reflects the more distant location of water in contact to the PEM film (S compartment) compared to the bare Si-IRE (R compartment) with respect to the Si surface. Hence the $\nu(OH)$ band senses film formation as the removal of water from the surface in

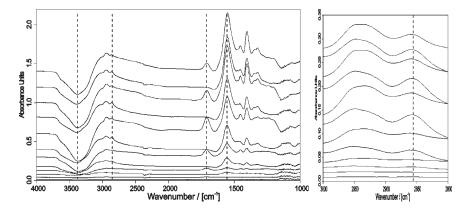


Figure 5. Left: Original in-situ-ATR-FTIR spectra (uncorrected data, see Experimental) on the consecutive deposition of PEM from solutions of PEI at pH \approx 10 and PAC at pH \approx 4 for c_{PEL} = 0.005 M and t_{ADS} = 5 min onto Si-IRE. PEM-z are shown from z = 1 to 12 from bottom to top. Right: respective extended ν (CH) spectral regions (PEI). Typical IR bands used for further analysis are indicated by vertical broken lines (see text).

Table 1.Assignment of IR bands and lineshape components appearing in ATR-FTIR spectra of PEM-PEI/PAC.

Wavenumber/[cm ⁻¹]	Assignment	Component
3380	ν(OH)	H₂O
2955	ν(CH)	PEI
1707	$\nu(C=O)$	PAC
1643	δ (OH)	H ₂ O
1552	$\nu_{a}(COO^{-})$	PAC

dependence of the adsorption step z. A summary of selected IR bands or the fitted band components (see above) used in this study and their assignment is given in the Table 1.

Deposition Profiles

The integrated areas of the selected IR bands were used to quantify PEM-PEI/PAC deposition. Most relevantly, the $\nu(\text{CH})$ band at 2855 cm⁻¹ (A_{CH}) can be used as a direct diagnostic measure for the deposited amount of PEI, while for the deposited amount of PAC the fitted band components $\nu(\text{C=O})$ (A_{C=O}) and

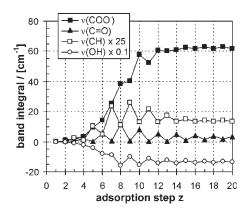


Figure 6.

Courses of the integrals of the ν (CH) band at 2855 cm $^{-1}$ (white cubes) assigned to PEI, the fitted ν (C=O) and ν (COO $^-$) band components at 1707 and 1552 cm $^{-1}$ (black triangle, cube) assigned to PAC and the ν (OH) band (white circle) of water versus z measured during consecutive PEM-z deposition from PEI/PAC solutions for pH \approx 10/4, c_{PEL} = 0.005 M and t_{ADS} = 5 min onto Si-IRE. The given band integrals are based on ATR-FTIR sprectra, which were corrected for the negative δ (OH) band at 1640 cm $^{-1}$ (see Experimental).

 $\nu_a({\rm COO^-})~({\rm A}_{{\rm COO^-}})$ have to be considered. In the Figure 6 typical courses of the integrals of $\nu({\rm COO^-}), \nu({\rm C=O}), \nu({\rm CH})$ and $\nu({\rm OH})$ band versus the adsorption step z are given, which are discussed in the following:

$v(COO^{-})$ Band Component

The course of the $\nu(COO^-)$ band component (A_{COO}-) due to PAC shows after a certain quasi linear lag phase until around z=4 a steep rise and an asymptotic behaviour for the adsorption steps z = 5-20. This is primarily due to the ATR-FTIR mode, which senses lateral PEM deposition as a linear function of A versus z, while vertical PEM deposition is sensed as an exponentially damped function given in Equation (1) and drawn in the Figure 3 (Experimental section). Hence, we conclude a short initial phase of mainly lateral growth up to z=4, followed by a mainly vertical PEM deposition from z = 5-20.

v(C=O) Band Component

For the $\nu(C=O)$ band components $(A_{C=O})$ due to PAC a non trivial course was found. In the modulated course of $A_{C=O}$, the values for even steps were always higher compared to the odd step before and after. This is generally due to a periodic uptake and loss of carboxylic acid (COOH) groups. Thereby the COOH group decrease in the odd steps can be caused either by deprotonation or also by the loss of PAC molecules. Since COOH and COO⁻ groups are complementary, the deprotonation of COOH should result in a gain of COOgroups. However, since e.g. the decrease of $A_{C=0}$ from z = 8 to z = 9 is not paralleled by a corresponding rise of A_{COO^-} , we conclude, that predominantly protonated PAC is released in every PEI step. Hence from both modulated courses PEM loss tendencies were found, when either PEI or PAC is immersed on the actual PEM-z to create PEM-(z+1). Furthermore $A_{C=O}$ shows a transient behaviour with a maximum at z=8, which is due to the ATR-FTIR sensing characteristics: i.e. at z = 8 there is a maximum of COOH groups sensed, while for subsequent steps z = 10, 12-20, even assuming a similar number of COOH groups in the outermost region, a smaller A_{C=O} value was measured. From that we conclude on the one hand a considerable accumulation of PAC at the outermost PEM surface region. On the other hand for z=8 the roughness of the PEM film could enable the PAC solution to contact PEM regions of small thickness or even the bare silicon surface. so that the evanescent field for PEM-8 senses PAC stronger compared to PEM-20, where accumulated PAC is no longer near the silicon surface or in the region of stronger evanescent field (see also $\nu(CH)$ band below).

ν(CH) Band

For the band integrals of ν (CH) due to PEI (further denoted as A_{PEI}) also a non trivial course was found. First of all strong modulation features can be identified, so that for PEI in every odd step A_{CH} was higher compared to the even step before and after. This modulated course of A_{CH} is due to a periodic uptake of PEI material for the odd PEI steps z = 1-19 and a respective loss of PEI material for the even PAC steps z = 2-20. From this we conclude that the initially bound PEI is partly released by the oppositely charged PAC into solution. Secondly very interestingly, in the Figure 6 the maximum A_{CH} value for z = 9 $(\approx 26 \text{ cm}^{-1})$ is higher compared to the saturation value (\approx 14 cm⁻¹) at z = 20 by a factor of nearly 2. Hence, considering a thickness increment from 0 to 500 nm, which is about in the range of the penetration depth (see Experimental), a higher PEI amount can be concluded for thinner PEM-9 at 0.005 M, where mainly the outermost bound PEI amount is sensed, compared to thicker PEM-19, where mainly the internally bound PEI amount is sensed. Analogously to the findings for A_{PAC} , this means that in the outermost regions of e.g. PEM-19 a similar large PEI uptake may take place, but within the given thickness increment this is no longer detected by

ATR-FTIR. At that point it has to be noted, that the difference of the refractive indices for water ($n_3 = 1.33$) dominating for PEM-9 and for the PEM film $(n_2 = 1.5)$ dominating for PEM-20 could not cause this absorbance difference. According to Harrick^[29] an increase in n₂ should rather result in a higher A_{CH} value by some factor of 1.09. These results can be interpreted based on two models, which may interfere somehow. On the one hand the outermost region of this PEM type may always consist of a huge amount of accumulated excess PEI, which is exceeding the internally bound amount by far. On the other hand the PEM may always be in a highly eroded state, so that until a certain z the PEI solution may still contact regions of bare silicon substrate sensed maximum by ATR-FTIR. The first interpretation would be in line with an accepted three zone model,[17,31] where the outermost zone is claimed to be highly nonstoichiometric, while in the internal bulk zone 1:1 stoichiometry prevails. In the respective subsequent steps this excess PEL amount is desorbed to a high degree, which can be seen by the huge modulation amplitude for PEI/PAC. However, the true picture accounts additionally for highly eroded PEM (second interpretation), where the three zone model is only locally valid. This point will be addressed in the morphology part below.

ν(OH) Band

Finally, in the Figure 6 also the modulated course of the negative $\nu(OH)$ band is given, which reflects the location of water with respect to the bare silicon surface or the sensing evanescent field. Significantly, in the odd PEI steps the integral of the $\nu(OH)$ band (A_{H2O}) is always higher compared to the even PAC steps before and after. This tendency can be due to two contributions: Either outermost PEI layers are more hydratable compared to outermost PAC layers, or in the PEI steps the thickness or deposited amount of the whole PEM is always smaller compared to the PAC step before and after. A detailed interpretation will be published therein.^[30]

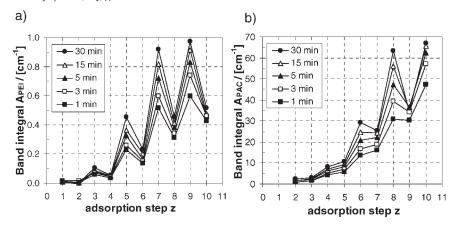


Figure 7. a) A_{PEI} of PEM-z plotted versus adsorption step z=1 to 10 for $t_{ADS}=0.5$, 1, 3, 5, 15, 30 min (indicated). b) A_{PAC} of PEM-z plotted versus adsorption step z=1 to 10 for $t_{ADS}=0.5$, 1, 3, 5, 15, 30 min (indicated).

Influence of Adsorption Time t_{ADS}

Furthermore, PEM deposition was studied varying the adsorption time t_{ADS} . In the Figure 7a the courses of A_{PEI} and in the Figure 7b those of A_{PAC} are shown in dependence of the adsorption step for various t_{ADS} values related to the individual adsorption steps. A_{PAC} was calculated according to $A_{PAC} = F \cdot A_{C=O} + A_{COO}$ (see Experimental) to address the total PAC amount. The A_{PEI} and A_{PAC} courses reflect the actual uptake and release of PEI and PAC, respectively, at the support surface. In both sets of deposition profiles a significant increase of both the overall

deposited amount and of the uptake/release amplitudes in dependence of $t_{\rm ADS}$ is seen. These uptake/release tendencies for PEI and PAC are in line with results from studies of Cohen-Stuart and coworkers [18], who reported similar tendencies for a different system. As it was also claimed therein, obviously, the PEM deposition process is kinetically controlled.

As a summary, Figure 8a shows the direct dependence of the band integrals A_{PEI} and A_{PAC} measured for PEM-6 on t_{ADS} . The courses take initially a steep ascent until $t_{ADS} \approx 5$ min, followed by a less steep one. Practically, this means

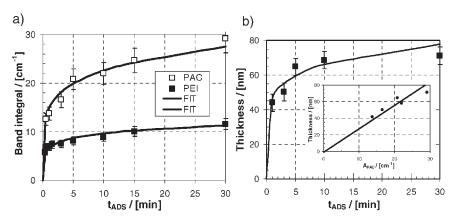


Figure 8. a) A_{PEI} (\times 50) and A_{PAC} of PEM-6 plotted versus t_{ADS} . The full lines are fitting curves using Equation (4). b) Thickness values of PEM-6 (SFM line cut method) plotted versus t_{ADS} (full lines: Fit by Equation (4)). Inlet: thickness (SFM plotted versus A_{PAC} (ATR-FTIR).

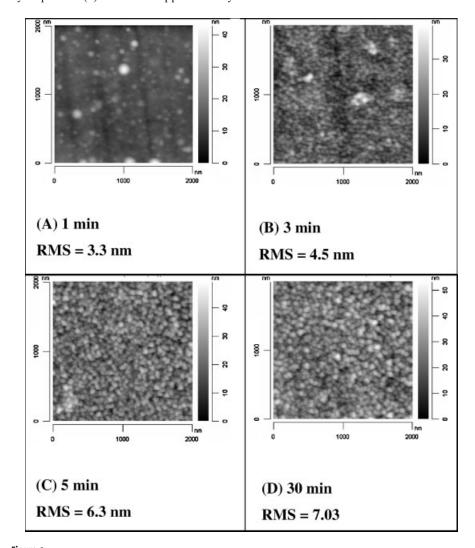
that $t_{\rm ADS} \approx 5$ min is sufficient for that PEM type under the chosen pH and $c_{\rm PEL}$ conditions to obtain an effective PEM deposition. Empirically, in the given range of $t_{\rm ADS} = 0.33$ –30 min the deposited amount scales with $t_{\rm ADS}$ by an empirical Smoluchowski like function (4) suggesting an aggregation related deposition process:

$$A = A_0 t_{ADS} b (4)$$

Fitting the experimental data of both band integrals A_{PEI} and A_{PAC} empirically by Equation (4) results in approximately

similar b values for PEI ($b=0.15\pm0.02$) and PAC ($b=0.18\pm0.01$). The fitted curves are shown as black lines in Figure 8a. From that generally an aggregation type of deposition process might be speculated on and similar incorporation rates of PEI and PAC into the PEM might be concluded.

The results of Figure 8a obtained by ATR-FTIR were checked with the SFM line cut method as another sensitive analytical method delivering thickness values of the PEM films. For that the same



Typical SFM images (topography) on PEM-6 samples deposited under constant pH setting of 10/4 (PEI/PAC) and $c_{\text{PEL}} = 0.005$ M for various adsorption times t_{ADS} .

PEM-6 samples as they were used in the ATR-FTIR analysis were considered. The results are shown in the Figure 8b. Significantly, a similar trend of PEM deposition in dependence of tADS was found: after an additional steep rise of the thickness until $t_{ADS} = 5$ min the further thickness increase with t_{ADS} is only low. Hence, practically the PEM deposition process for PEI/PAC at cPEL = 0.005 M does not alter that much applying t_{ADS} 5 min. Hence, this value seems to be sufficient for an effective and quick process. Fitting the thickness data with Equation (4) replacing A and A₀ by d and d₀ led to $b = 0.15 \pm 0.03$, which confirms the ATR-FTIR data (the fitted curve is shown as full line). From this result it could be additionally concluded, that if thicknesses within the range 0–300 nm are studied, the band integrals obtained by ATR-FTIR analysis are approximately linear with the "true" thickness values obtained by SFM line cut method. This observation is supported by the inlet in the Figure 8b, where the thickness found by SFM line cut method is plotted versus the A_{PAC} values of the ATR analysis. From this plot an approximate linear relationship between the observables of both methods was confirmed, which seems to be the more exact the lower the thicknesses were.

PEM Nanostructure

The equivalent PEM samples investigated by ATR-FTIR spectroscopy were studied by scanning force microscopy (SFM) aiming at the surface morphology and roughness in depen-dence of $t_{\rm ADS}$. For that purpose only PEM-6 samples deposited under constant pH (10/4) and $c_{\rm PEL}$ (0.005 M) were considered. Four PEM-6 samples were chosen deposited during $t_{\rm ADS} = 1, 3, 5$ and 30 min, and in the Figure 9 A, B, C typical SFM images of these samples are shown.

There the structures and especially their sizes were slightly changing in dependence of $t_{\rm ADS}$. For $t_{\rm ADS} = 1$ min (A) individually appearing dots on a relative smooth base layer were found, which changed to more continuous layers of crowded dots for

 $t_{\rm ADS}\!=\!3$ min (B). For $t_{\rm ADS}\!=\!5$ and 30 min (A, D) the dots were more densely packed with a tendency to merging. Generally an enlargement of the surface structure in dependence of $t_{\rm ADS}$ could be derived. Included in the Figure 9 are roughness (RMS) values, which increase with increasing $t_{\rm ADS}$ and correlate qualitatively with the found deposited amounts given in Figure 8b. Presumably, the more PEM grow in the vertical direction with increasing adsorption time, the more roughness is created.

Conclusion

Quantitative ATR-FTIR spectroscopy and SFM were applied on deposition of PEM of PEI and PAC in dependence of adsorption time (t_{ADS}) at fixed pH and PEL concentration c_{PEL} providing for selective informations on PEI (ν (CH)) and PAC (ν (COO⁻), ν (C=O)) uptake and release, respectively, as well as on surface morphology.

Plotting band integrals of these bands versus t_{ADS} showed, that the deposited amount of PEI and PAC increased steeply between $t_{ADS} = 0$ to 5 min followed by a small further increase for $t_{ADS} > 5$ min. This was confirmed by SFM profile measurements on line cuts through the PEM films providing for the PEM thickness (d_{SFM}). Within a thickness range from 0–100 nm the band integrals related to PEI and PAC (ATR-FTIR) were shown to be approximately linear with d_{SFM} (this would be not the case for thicknesses exceeding the ATR penetration depth (see Experimental)). An empirical Smoluchowski like analytical function represented the data quite sufficiently, suggesting an aggregation type of deposition process.

From oscillations features in the deposition profiles, the release of PEI upon PAC immersion (even steps) and of PAC upon PEI immersion (odd steps) could be claimed for all applied $t_{\rm ADS}$. From that a competition between PEL surface uptake and PEL solution complexation formation

can be concluded in line with studies of Cohen-Stuart and coworkers.^[18]

SFM data on PEM-6 samples deposited at constant pH = 10/4 and $c_{\rm PEL} = 0.005$ M in dependence of $t_{\rm ADS}$ showed isolated structures for $t_{\rm ADS} = 1$ min, moderately fused ones for $t_{\rm ADS} = 3$ min and further fused ones for $t_{\rm ADS} \ge 5$ min. Additionally, a subsequent roughness increase from around 3 to 7 nm was found.

These studies help to prepare polyelectrolyte based films with controlled thickness for the interaction with biofluids in the biomedical and food field.

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